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SEMICONDUCTOR AND FABRICATION METHOD THEREOF

BACKGROUND OF THE INVENTION

The present invention relates to a semiconductor device and a fabrication method thereof, and particularly to a semiconductor device in which an n-channel field effect transistor and a p-channel field effect transistor are provided on a common base-substrate, and a fabrication method thereof.

In recent years, along with rapid advance in digitization of information processing means, there have been strong demands to increase the operational speeds of semiconductor devices and to reduce the power consumption of semiconductor devices. Conventionally, the increase in operational speed of semiconductor devices has been achieved by making finer device structures, and the reduction in power consumption thereof has been achieved by adopting semiconductor devices of a complementary type including n-channel and p-channel field effect transistors. The technique for making finer device structures, however, has already reached a limitation of a lithography process, in which a pattern having a size smaller than a wavelength of exposure light has been needed to be formed by lithography, and accordingly, at

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the lithography step, it has come to be difficult to ensure a sufficient process tolerance. In other words, the attempt to increase the operational speeds of semiconductor devices by making finer device structures is coming closer to a limitation.

For field effect transistors, there has been proposed an attempt to solve the above problem by imparting a strain effect to a channel layer. The formation of devices using a material layer having such a strain effect has become possible by an advanced thin film formation technology using a group IV element containing semiconductor material such as silicon or a silicon-germanium compound, and at present, improved low voltage devices have been actively developed by using material layers having the strain effect.

The strain effect means a phenomenon that if stress is applied to a semiconductor thin film, the energy band of the semiconductor is strained, to thereby change the effective masses of carriers in the semiconductor. The formation of a semiconductor thin film having the strain effect has become possible by controlling an internal stress applied to a multi-layer film typically containing a silicon layer and a silicon-germanium compound layer by a molecular beam epitaxy process or an ultrahigh vacuum

chemical vapor deposition process (UHV-CVD). High performance MOS based devices, sensors, and the like of a type in which a difference in band gap and film strain are controlled by hetero-junction have been also developed.

A silicon based MOS (Metal-Oxide-Semiconductor) including layers each having a strain effect can be produced by sequentially stacking a buffer layer made from a silicon-germanium compound having a germanium concentration gradually increased toward an upper surface of the buffer layer, a relax layer made from a silicon-germanium compound, and a silicon layer on a silicon substrate in this order. In this MOS transistor, since tensile stress is generated in the silicon layer, a strain effect due to the tensile stress appears, with a result that the mobility of electrons is increased by the strain effect in the silicon layer; and since compressive stress is generated in the silicon-germanium compound layer formed on the silicon substrate, a strain effect due to the compressive stress appears, with a result that the mobility of positive holes is increased by the strain effect in the silicon-germanium compound layer.

A field effect transistor produced by controlling stress in a channel layer by making use of the above-

described strain effect exhibits a high mutual conductance $[gm \text{ (mobility)}]$. A pMOS transistor produced by making use of the strain effect has been disclosed in Appl. Phys. Letter (USA), 63(1993) S. P. Voinigensen et al., p660, and IEEE Electronic Devices (USA), 43(1996), L. H. Jiang and R. G. Elliman, p97. Further, an nMOS transistor produced by making use of the strain effect has been disclosed in Appl. Phys. Letter (USA), 64(1994) K. Ismail et al., p3124 and IEDM 94-37 (USA), (1994) J. Welser et al.

The semiconductor devices produced by making use of the strain effect have the following problem:

In a silicon layer formed on a relax layer, the mobility of electrons is increased by the strain effect due to tensile stress; however, the mobility of positive holes is reduced. Accordingly, while an nMOS transistor having an improved low voltage can be obtained, the performance of a pMOS transistor cannot be improved, with a result that the performance of the CMOS cannot be significantly improved.

On the other hand, in a silicon-germanium compound layer formed on a silicon substrate, the mobility of positive holes is improved by the strain effect due to compressive stress; however, the mobility of electrons is

reduced. Accordingly, while a pMOS transistor having an improved low voltage can be obtained, the performance of an nMOS transistor cannot be improved, with a result that the performance of the CMOS cannot be significantly improved.

As described above, according to the related art method, it has been impossible to obtain a high performance CMOS and hence to produce a semiconductor device having a high performance and a low power consumption.

SUMMARY OF THE INVENTION

An object of the present invention is to provide a semiconductor device capable of increasing the operational speed and reducing the power consumption by providing an nMOS in which the mobility of electrons is increased by a strain effect and a pMOS in which the mobility of positive holes is increased by the strain effect on a common base-substrate, and to provide a method of fabricating the semiconductor device by using the same fabrication process as that used for fabricating a related art CMOS in which the base-substrate is configured as only a silicon substrate.

To achieve the above object, according to a first

regions of the p-channel field effect transistor may be formed in the silicon-germanium compound layer.

With this configuration, since the semiconductor device has the CMOS configuration in which the n-channel field effect transistor and the p-channel field effect transistor are provided on the common base-substrate, the power consumption can be reduced. Further, in the n-channel field effect transistor, since the silicon layer is provided on the relax layer made from the silicon-germanium compound whose stress is relaxed because it is formed on the buffer layer, tensile stress is generated in the silicon layer, so that the mobility of electrons is increased by the strain effect due to tensile stress in the silicon layer; and in the p-channel field effect transistor, since the silicon-germanium compound layer is provided on the silicon substrate, compressive stress is generated in the silicon-germanium compound layer, so that the mobility of positive holes is increased due to the strain effect due to the compressive stress in the silicon-germanium compound layer. According to the semiconductor device of the present invention, since the source/drain regions are formed in each of the silicon layer and silicon-germanium compound layer having the above-described unique properties, the operational speeds

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of the n-channel field effect transistor and p-channel field transistor can be improved.

According to a second aspect of the present invention, there is provided a method of fabricating a semiconductor device in which an n-channel field effect transistor and a p-channel field effect transistor are provided on a common base-substrate, the method including the steps of: etching a surface layer of an n-channel region, in which the n-channel field effect transistor is to be formed, of a silicon substrate, to form a stepped recess at the n-type region; forming a buffer layer made from a silicon-germanium compound having a germanium concentration gradually increased toward an upper surface of the buffer layer on the silicon substrate; forming a relax layer made from a silicon-germanium compound having a germanium concentration nearly equal to that of a surface portion of the buffer layer on the buffer layer; forming a silicon layer on the relax layer; removing the silicon layer, the relax layer, and the buffer layer in a p-type region, on which a p-channel field effect transistor is to be formed, of the silicon substrate, by etching using a resist pattern as a mask; forming a silicon-germanium compound layer on both the silicon substrate and the silicon layer; planarizing the surface

of the silicon-germanium compound layer in a state that the silicon layer is exposed in the n-type region and the silicon-germanium compound layer is left in the p-type region; forming a silicon epitaxial layer on both the silicon layer and the silicon-germanium compound layer, to form a silicon layer composed of the silicon layer and the silicon epitaxial layer in the n-type region, and to form a cap layer composed of the silicon epitaxial layer in the p-type region; and forming gate electrodes on the silicon layer in the n-type region and on the cap layer in the p-type region via a gate insulating film, and forming n-type source/drain regions in the silicon layer and p-type source/drain regions in the silicon-germanium compound layer.

With this configuration, the n-channel field effect transistor, in which the silicon layer is formed on the silicon substrate via the buffer layer and relax layer and the source/drain regions are formed in the silicon layer, is provided in the n-type region; and a p-channel field effect transistor, in which a silicon-germanium compound layer is formed on the silicon substrate and the source/drain regions are formed in the silicon-germanium compound layer, is provided in the p-type region. In this case, the silicon-germanium layer is formed in the state

that the buffer layer, relax layer, and silicon layer are left in the stepped recess formed at the n-type region of the silicon substrate, and the surface of the silicon-germanium compound layer is planarized by a CMP process in such a manner as to expose the silicon layer only in the n-type region, and accordingly, the gate electrodes of the n-channel field effect transistor and p-channel field effect transistor are formed on a planar base-substrate. As a result, the planar structure of the semiconductor device of the present invention becomes similar to that of a related art semiconductor device in which the base-substrate is configured as only a silicon substrate.

BRIEF DESCRIPTION OF THE DRAWINGS

Fig. 1 is a sectional view of a semiconductor device according to the present invention; and

Figs. 2A to 2D are sectional views of fragmental structures of the semiconductor device shown in Fig. 1, for illustrating steps of fabricating the semiconductor device;

Figs. 3A to 3C are views similar to Figs. 2A to 2D showing the steps, continued from those shown in Figs. 2A to 2D, of fabricating the semiconductor device;

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The buffer layer 2 is formed on the silicon substrate 1 by an ultrahigh vacuum chemical vapor deposition (UHV-CVD) process using a CVD system typically including a load lock chamber, a reaction chamber composed of a quartz tube, and an unload lock chamber. To be more specific, a quartz boat on which 20 pieces of upright silicon substrates 1 are placed is set in the reaction chamber, and the buffer layer 2 is formed on the silicon substrates 1. At this time, the substrate temperature is set in a range of 600 to 900°C, and a silane gas (SiH_4) and a germane gas (GeH_4) are made to flow in the reaction chamber in a state that hydrogen gas (H_2) or argon gas (Ar) has flown in the reaction chamber as a carrier gas. At the initial stage of film formation, only the carrier gas and the silane gas is made to flow, to deposit crystals of Si to a thickness of about 0.1 to 0.3 μm , thereby moderating the surface roughness of the silicon substrate 1 caused by etching. Subsequently, the germane gas is made to flow in the reaction chamber while adjusting the flow rate in accordance with a specific composition ratio of germanium. In this case, the flow rates of the silane gas and germane gas are adjusted by alternately introducing the silane gas and germane gas at

time intervals of a value typically ranging from several hundreds msec to several sec. This facilitates the adjustment of the flow rates of the silane gas and germane gas and improves the crystallinity of the buffer layer 2. The above value of time intervals is suitably selected depending on a conductance of each of the reaction chamber and evacuation system and a gas residence time, and the flow rates of the silane gas and germane gas are adjusted by a computer-controlled highly responsive mass flow controller. Additionally, a single-wafer type CVD system may be used for formation of the buffer layer 2.

It is checked on off-line whether or not the buffer layer 2 thus formed has the specific composition, by analysis such as X-ray diffraction, Rutherford back-scattering (RBS), secondary ion mass spectrometry (SIMS), observation by transmission electron microscope (TEM), spectroscopic ellipsometry, or surface reflection.

Referring to Fig. 2C, a relax layer 3 having a thickness of about $0.6 \mu\text{m}$, which is made from a silicon-germanium compound ($\text{Si}_{1-y}\text{Ge}_y$), is formed on the buffer layer 2. The composition ratio y of germanium in the relax layer 3 is set at a value nearly equal to the composition ratio x of germanium at the uppermost portion

of the buffer layer 2. The relax layer 3 functions as a stress-relaxed layer.

Following the formation of the buffer layer 2, the relax layer 3 is formed in the above-described reaction chamber. Like the formation of the buffer layer 2, the silane gas and germane gas are alternatively introduced at time intervals of a value typically ranging from several hundreds msec to several sec.

Referring to Fig. 2D, the substrate temperature is cooled to 600°C in a state that a slight amount of the carrier gas is made to flow in the reaction chamber. In this case, to prevent occurrence of thermal stress due to rapid temperature drop, the substrate temperature is dropped at a rate of about -2°C/min. After the substrate temperature is dropped to 600°C, the silane gas is made to flow in addition to the carrier gas, to form a silicon layer 4 on the relax layer 3 by epitaxial growth. At this formation step, tensile stress is generated in the silicon layer 4. Accordingly, the silicon layer 4 functions as a layer having a strain effect due to the tensile stress generated in the layer. The silicon layer 4 also functions to protect the relax layer 3 from an external environment.

In addition, the thickness of the silicon layer 4

is set such that the surface of the n-type region 1a of the silicon layer is about 50-100 nm higher than the surface of the silicon substrate 1 in the p-type region.

Referring to Fig. 3A, a sacrifice oxide film 5 is formed on the silicon layer 4 by a thermal oxidation process or a CVD process. In this case, to suppress deterioration of the crystal state of each layer made from the silicon-germanium compound formed at the previous steps, the film formation temperature is set at a value of 850°C or less. In the case of forming the sacrifice oxide film 5 by rapid thermal anneal (RTA) or laser anneal, however, the film formation may be performed under a condition that the heating temperature at a limited region is suppressed at a value less than 937°C which is a melting point of germanium.

Referring to Fig. 3B, the sacrifice oxide film 5, silicon layer 4, relax layer 3, and buffer layer 2 in the p-type region 1b are removed by etching using a resist pattern (not shown) having a thickness of about 2.3 μm as a mask, to expose the silicon substrate 1 in the p-type region 1b. The resist pattern has an opening portion corresponding to the p-type region 1b and a mask portion corresponding to the n-type region 1a. The etching is performed in high density plasma containing a fluorine

based gas such as methane tetrafluoride (CF_4). The resist pattern is removed after etching.

The silicon substrate 1 is overall subjected to chemical mechanical polishing (CMP), to remove damages, caused by etching, on the surface layer of the silicon substrate 1. At this CMP step, only the surface layer of the silicon substrate 1 in the p-type region 1b is polished because the sacrifice oxide film 5 functions as a mask for the n-type region 1a.

Referring to Fig. 3C, the sacrifice oxide film 5 is removed by wet etching using dilute hydrofluoric acid. After removal of the sacrifice oxide film 5, the surface of the silicon layer 4 in the n-type region 1a is about 50-100 nm higher than the surface of the silicon substrate 1 in the p-type region 1b.

Referring to Fig. 4A, a silicon-germanium compound ($\text{Si}_{1-z}\text{Ge}_z$) layer 6 having a thickness of about 100 nm is formed on both the silicon substrate 1 and the silicon layer 4. The composition ratio z of germanium in the silicon-germanium compound is set in a range of 0.1 to 0.8, preferably, at a value near 0.2. Like the formation of the buffer layer 2, the formation of the silicon-germanium compound layer 6 is performed by alternately introducing the silane gas and germanium gas at time

intervals of a value ranging from several hundreds msec to several sec. The silicon-germanium compound layer 6 formed on the silicon substrate 1 functions as a layer having a strain effect due to compressive stress generated in the layer.

Referring to Fig. 4B, the surface of the silicon-germanium compound layer 6 is planarized by the CMP process, to expose the silicon layer 4 in the n-type region 1a while leaving the silicon-germanium compound layer 6 in the p-type region 1b. With this planarization, the thickness of the silicon layer 4 in the n-type region 1a becomes about 20 nm, and the thickness of the silicon-germanium compound layer 6 in the p-type region 1b becomes about 10 nm.

Referring to Fig. 4C, a silicon epitaxial layer 7 having a thickness of about 100 nm is formed on both the silicon layer 4 and the silicon-germanium compound layer 6, whereby a silicon layer 7a composed of the silicon epitaxial layer 7 and the silicon layer 4 is formed in the n-type region 1a, and the silicon epitaxial layer 7 becomes a cap layer 7b in the p-type region 1b. In the case of formation of a gate insulating film at the subsequent step, the cap layer 7b functions to improve the quality of the gate insulating film.

polysilicon film 12 on the silicon layer 7a in the n-type region 1a and the cap layer 7b in the p-type region 1b via the gate oxide film 11.

After a resist-pattern covering the p-type region 1b is set, ions of an n-type impurity are implanted in the silicon layer 7a in the n-type region 1a by using the gate electrode 12a as a mask, to form source/drain regions 13a in the silicon layer 7a. Similarly, after a resist-pattern covering the n-type region 1a is set, ions of a p-type impurity are implanted in the silicon-germanium compound layer 6 in the p-type region 1b by using the gate electrode 12a as a mask, to form source/drains regions 13b in the silicon-germanium compound layer 6. The resist pattern is removed after ion implantation.

Subsequently, side walls 14 made from an insulating film are formed on side walls of each of the gate electrodes 12a and gate oxide film portions 11.

In this way, a semiconductor device, in which the nMOSTr 15a is provided in the n-type region 1a of the silicon substrate 1 and the pMOSTr 15b is provided in the p-type region 1b of the same silicon substrate 1 is obtained. In the nMOSTr 15a of the semiconductor device thus formed, the source/drain region 13a are formed in

the silicon layer 7a formed on the silicon substrate 1 via the buffer layer 2 made from the silicon-germanium compound and the relax layer 3 made from the silicon-germanium compound layer, and in the pMOSTr 15b of the semiconductor device, the source/drain regions 13b are formed in the silicon-germanium compound layer 6 formed on the silicon substrate 1.

In the nMOSTr 15a, since the silicon layer 7a is provided on the relax layer 3 made from the silicon-germanium compound whose stress is relaxed because it is formed on the buffer layer 2, tensile stress is generated in the silicon layer 7a, so that the mobility of electrons is increased by the strain effect due to tensile stress in the silicon layer 7a. As a result, the operational speed of the nMOSTr 15a in which the source/drains regions 13a are formed in the silicon layer 7a is improved.

On the other hand, in the pMOSTr 15b, since the silicon-germanium compound layer 6 is provided on the silicon substrate 1, compressive stress is generated in the silicon-germanium compound layer 6, so that the mobility of positive holes is increased due to the strain effect due to the compressive stress in the silicon-germanium compound layer 6. As a result, the operational

speed of the pMOSTr 15b in which the source/drain regions 13b are formed in the silicon-germanium compound layer 6 is improved.

Another advantage of the semiconductor device is that since the semiconductor device has a CMOS configuration that the nMOSTr 15a and the pMOSTr 15b are provided on the common base-substrate, the power consumption thereof can be reduced. The semiconductor device has a further advantage that since the operational speed of each of the nMOSTr 15a and pMOSTr 15b is improved, the operational speed of a logic circuit can be made as high as several times to several tens times as compared with the related art CMOS in which the base-substrate is configured as only the silicon substrate.

According to this embodiment, since a portion in which crystal defects are easy to occur, that is, a boundary portion between the n-type region 1a and the p-type region 1b is removed at the trench isolation 10a formation step described with respect to the Figs. 4A and 4B, it is possible to obtain a desired transistor performance.

Since the planar structure of the semiconductor device in this embodiment is similar to the related art CMOS in which the base-substrate is configured as only

